



# Synthesis and Characterization of Indium Oxide Nanoparticles Using Indium Nitrate and Polyvinylpyrrolidone (PVP) as Precursors

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## ABSTRACT

This paper reports the synthesis and characterization of  $\text{In}_2\text{O}_3$  nanoparticles by a simple polymerized complex method using indium (III) nitrate and polyvinylpyrrolidone (PVP) as the starting chemicals. The precursors were calcined at different temperatures to obtain  $\text{In}_2\text{O}_3$  nanoparticles. The XRD and electron diffraction analysis results indicated that the  $\text{In}_2\text{O}_3$  samples had a cubic structure without any impurity phases. The FT-IR analysis result confirmed the formation of  $\text{In}_2\text{O}_3$ . TEM revealed that the  $\text{In}_2\text{O}_3$  samples consist of crystalline particle of 15-25 nm which are weakly agglomerated. UV-vis spectroscopy was also performed to study the optical properties of synthesized  $\text{In}_2\text{O}_3$  samples. The estimated band gap of the samples was 3.70-3.93 eV.

**Keywords:** indium oxide, nanoparticles, polyvinylpyrrolidone (PVP), cubic structure and transmission electron microscopy

## 1. INTRODUCTION

Indium oxide ( $\text{In}_2\text{O}_3$ ) is a wide band gap n-type semiconductor with a direct band gap of 3.75 eV.  $\text{In}_2\text{O}_3$  is an insulator in a stoichiometric state, but behaves as a high conductive semiconductor in its non-stoichiometric state.  $\text{In}_2\text{O}_3$  has a cubic bixbyite structure with lattice parameter of 10.117 Å [1-2]. Apart from this,  $\text{In}_2\text{O}_3$  in its nanostructure form exhibits noteworthy applications and is currently gaining much attention. The reason

lies in the fact that nanomaterials have to exhibit many specific features in order to meet all the expectations that are related to their tiny size. Because physical and chemical properties of nanomaterials depend not only on the composition but also on the particle size and shape, a good synthesis protocol first of all needs to provide control over particle size and shape [3]. Seetha et al. [4] reported visible light emitting  $\text{In}_2\text{O}_3$  nanoparticles synthesized

by the precipitation method. Sharma et al. [5] reported that thin films of  $\text{In}_2\text{O}_3$  are deposited onto indium–tin oxide (ITO) coated glass substrates by the electrodeposition technique to enhance their optical properties. Huang et al. [6] prepared through the liquid precipitation method, an application for gas-sensing. Qin et al. [7] reported the fabrication and characterization of efficient organic solar cells (OSCs) based on p-type  $\text{In}_2\text{O}_3$  films as the hole-transporting layer (HTL) by the radio frequency (RF) magnetron sputtering method.

There are a variety of synthesis techniques (routes) for preparation of  $\text{In}_2\text{O}_3$  nanoparticles such as sol–gel [8, 9], chemical vapor deposition [10–11], hydrothermal [12–13] and electrospinning [14–16]. Recent studies on un-doped and doped  $\text{In}_2\text{O}_3$  have mainly concentrated on preparation of various nano-structural shapes and their properties. For practical applications, a significant interest has been shown in the field of zero-dimensional and one-dimensional semiconducting  $\text{In}_2\text{O}_3$  and indium tin oxide (ITO), such as nanospheres [5, 17], nanorods [8, 10], nanopyramids [18], nanofibers [14–16], nanowires [19] and nanoparticles [11, 19–21]. Among other established synthesis methods, simple and cost effective routes to synthesize nanocrystalline  $\text{In}_2\text{O}_3$  by utilization of cheap, nontoxic and environmentally benign precursors are still the key issues, as well as the ability to control particle size and shape. A simple method has been successfully used by our group to synthesize  $\text{CeO}_2$  nanoparticles with particle sizes of ~5–10 nm [22], nanocrystalline Zn and Co-doped ZnO powders with particle sizes of ~50–100 nm and nanorods with dimeters of ~100–200 nm [23–24].

In this paper, we investigated the applicability of polyvinyl pyrrolidone (PVP) in preparing  $\text{In}_2\text{O}_3$  nanoparticles using Indium (III) nitrate n-hydrate and PVP as the starting chemicals. The synthesized precursors were calcined at 450, 500, 550 and 600°C for 2h to obtain the

$\text{In}_2\text{O}_3$  nanoparticles. The synthesized powders were characterized by X-ray diffraction (XRD), transmission electron microscopy (TEM), ultraviolet-visible spectroscopy (UV-Vis) and fourier infrared spectroscopy (FT-IR).

## 2. MATERIALS AND METHODS

In this study, indium (III) nitrate n-hydrate (Kanto, 99.9%) and PVP ( $M_n=1300000$ , Aldrich) were used as the starting chemicals. In a typical procedure, 7.5 g of polyvinyl pyrrolidone (PVP) was first dissolved 500 ml distilled water under vigorous stirring at 50°C. Subsequently, 5 g of indium nitrate n-hydrate was slowly added to this solution. The transparent white-colored mixture was then stirred for several hours in air until dry. At this step, the PVP acted as a matrix for entrapment of indium ion and it generated a gelled precursor [22–24]. Throughout the whole process described above, no pH adjustment was made.

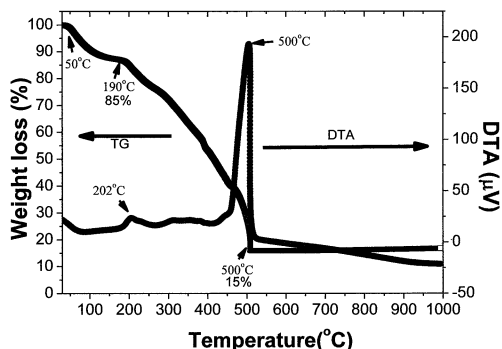
Then, the precursor was mixed well and a yellow powder can be obtained after 30 min of milling done in a mortar and pestle, by hand at room temperature. Finally, the dried precursor was calcined in a box furnace at 450, 500, 550 and 600°C for 2h in air. The calcined samples were ground to break up large agglomerates. The final products obtained were yellow white  $\text{In}_2\text{O}_3$  nanoparticles.

In order to determine the temperature of passible decomposition and crystallization of the powders, the dried powder precursor was subjected to TG-DTA (SSDT Q600, Luken's drive, Newcastle) at a heating rate of 10°C / min in air up to 1000°C. The prepared  $\text{In}_2\text{O}_3$  sample was characterized by X-ray diffraction at room temperature using a Philips X-ray diffractometer (PW3710, The Natherlands), working with  $\text{CuK}\alpha$  radiation in the  $2\theta$  range of 20°–90°. Fourier transform infrared (FT-IR) spectra of the powders (as pellets in KBr) were recorded using a fourier transform infrared spectrometer (PerkinElmer Instrument, England)

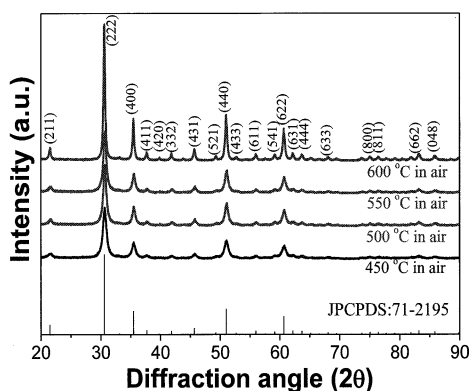
in the range of  $4000\text{--}400\text{ cm}^{-1}$  with a resolution of  $1\text{ cm}^{-1}$ . The particle size and morphology of the calcined powders were characterized by transmission electron microscopy (JEOL 2010, 200 kV, England). The optical absorption spectra were measured in the range of 200–800 nm using a UV-vis Spectrometer (T80/T80t, PG Instruments Limited, UK).

### 3. RESULTS AND DISCUSSION

The TG and DTA curves of the powder precursor of  $\text{In}_2\text{O}_3$  are shown in Figure 1. The TG curve in Figure 1 shows a major weight loss step from  $\sim 200^\circ\text{C}$  up to about  $\sim 500^\circ\text{C}$  with slight weight loss from  $\sim 50^\circ\text{C}$  to  $\sim 195^\circ\text{C}$ , and no further weight loss was observed above  $500^\circ\text{C}$ . The weight loss is related to the combustion of the organic matrix. Weight loss occurs gradually at temperatures above room temperature and reaches  $\sim 15\%$  around  $200^\circ\text{C}$  in Figure 1. This weight reduction is due to the loss of absorbed water (mainly from hydration of the indium nitrate used as the starting material) [13] and thermal decomposition of the surfactant polymer. The significant weight loss of  $\sim 70\%$  in Figure 1 between  $\sim 200$  to  $\sim 500^\circ\text{C}$  was attributed to the complete decomposition of  $\text{In}(\text{NO}_3)_3$  and the degradation of PVP, which involves both intra- and intermolecular transfer reactions. The



**Figure 1.** TG-DTA curves of the thermal decomposition of the precursor at a heating rate of  $10^\circ\text{C}/\text{min}$  in static air.



**Figure 2.** XRD patterns of  $\text{In}_2\text{O}_3$  nanoparticles calcined in air for 2 h at  $450^\circ\text{C}$ ,  $500^\circ\text{C}$ ,  $550^\circ\text{C}$  and  $600^\circ\text{C}$ .

exothermic peaks at  $500^\circ\text{C}$  in Figure 1 were the result of the oxidation of carbon and carbon monoxide released by the decomposition of PVP [16]. The structure of the  $\text{In}_2\text{O}_3$  sample was primary examined by XRD. The XRD patterns from the  $\text{In}_2\text{O}_3$  samples are shown in Figure 2. All of the detectable peaks (Figure 2) can be indexed as the  $\text{In}_2\text{O}_3$  cubic structure in the standard data (JCPDS:71-2195). The cubic lattice parameters  $a$ , calculated from the XRD spectra are 1.0099, 1.0092, 1.0122 and 1.0119 nm for  $\text{In}_2\text{O}_3$  samples calcined at 450, 500, 550 and  $600^\circ\text{C}$ , respectively. These values are close to those of lattice constant  $a = 1.0117$  nm in the standard data (JCPDS: 71-2195). The crystallite sizes ( $D$ ) of the powders were estimated from X-ray line broadening using Scherrer's equation [25].

$$D = \frac{0.89\lambda}{\beta \cos \theta} \quad (1)$$

where  $\lambda$  is the wavelength of the X-ray radiation,  $K$  is a constant taken as 0.89,  $\theta$  is the diffraction angle,  $\beta$  is the full width at half maximum (FWHM), and were determined to be 12.1, 13.7, 17.1 and 23.7 nm for  $\text{In}_2\text{O}_3$  samples calcined at 450, 500, 550 and  $600^\circ\text{C}$ , respectively. The crystallite sizes and lattice parameters of  $\text{In}_2\text{O}_3$  samples are summarized in Table 1.

**Table 1.** Summary of particle sizes from XRD and TEM, cubic lattice constant and the band gap ( $E_g$ ) of the nanocrystalline  $\text{In}_2\text{O}_3$  samples calcined in air at different temperatures for 2h.

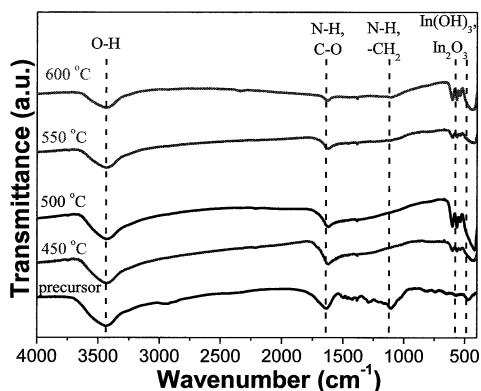
$\text{In}_2\text{O}_3$ sample	particle size (nm)		Cubic lattice parameter <sup>a</sup> a(nm)	Estimated band gap (eV)
	from XRD	from TEM		
Calcined at 450°C	12.1	15	1.0099	3.93
Calcined at 500°C	13.7	17	1.0092	3.90
Calcined at 550°C	17.1	19	1.0122	3.70
Calcined at 600°C	23.7	25	1.0119	3.82

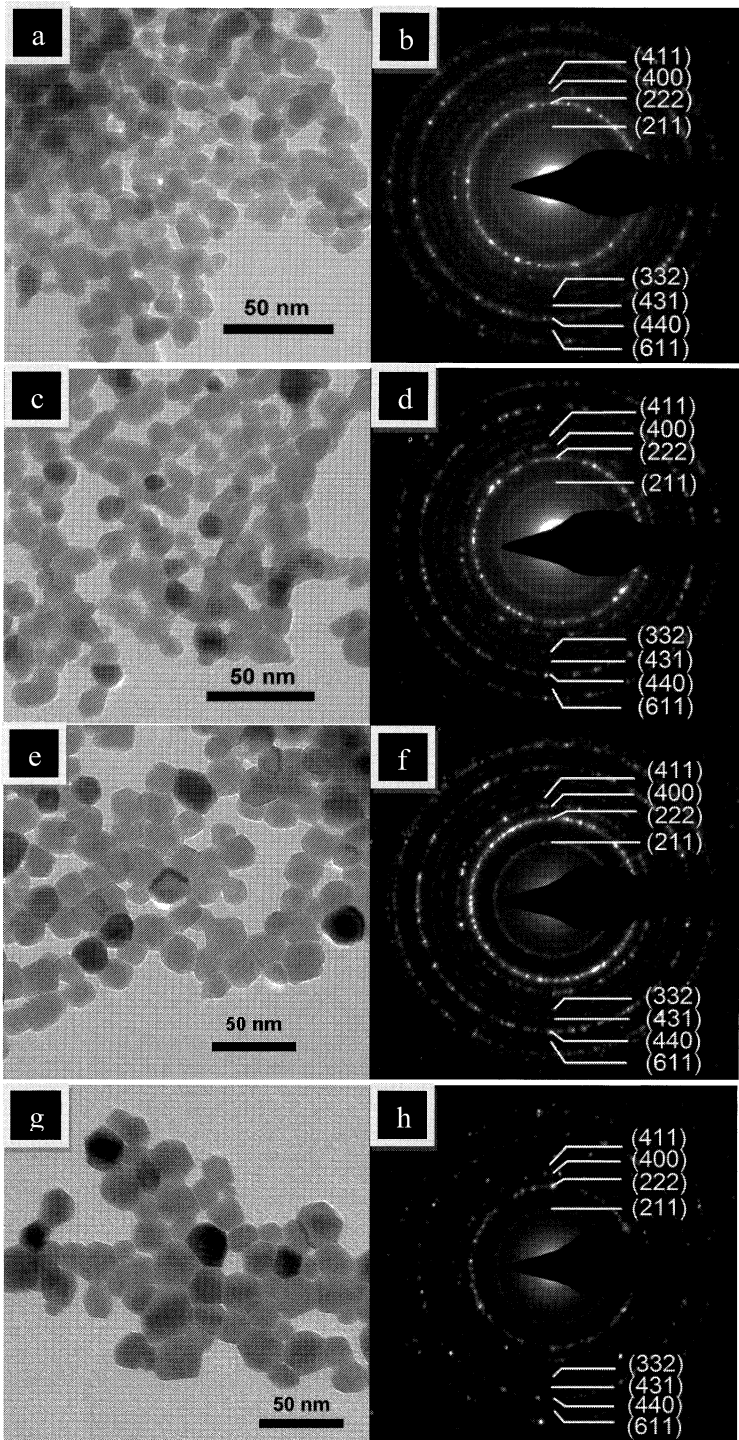
The formation of an  $\text{In}_2\text{O}_3$  bond was supported by FT-IR spectra as shown in Figure 3. In the FT-IR spectra absorption peaks ascribing to In-O bonds appear at wave numbers of  $\sim 475$ ,  $\sim 564$ , and  $\sim 645\text{ cm}^{-1}$ , suggesting that the phase transformation of  $\text{In}(\text{OH})_3$ -to-In-O occurs here, while C=O and N-H bonds with the PVP polymers are at  $\sim 1635\text{ cm}^{-1}$  and  $\sim 1109\text{ cm}^{-1}$  respectively [13]. These reveal that the PVP molecules adsorbed preferentially on the surface of  $\text{In}(\text{OH})_3$  crystallites. Disappearance of N-H and C-O bond from the PVP polymer at 500-600°C in Figure 3 indicates a complete removal of the polymer [22].

The morphology and structure of the  $\text{In}_2\text{O}_3$  samples were investigated by TEM. It is clear from the TEM bright-field images (Figure 4) that the morphology and size of  $\text{In}_2\text{O}_3$

nanoparticles were affected by the calcination temperature. The TEM bright-field images of  $\text{In}_2\text{O}_3$  show that the  $\text{In}_2\text{O}_3$  sample calcined at 450°C (Figure 4(a)) has nanoparticles with the smallest particle size of  $\sim 15\text{ nm}$  whereas at 500°C (Figure 4(c)) trains nanoparticles have sizes of  $\sim 17\text{ nm}$ . The  $\text{In}_2\text{O}_3$  sample calcined at 550°C (Figure 4(e)) consists of well dispersed particles of  $\sim 19\text{ nm}$  and the sample calcined at 600°C (Figure 4(g)) consists of particles of  $\sim 25\text{ nm}$  in diameter. The corresponding selected-area electron diffraction (SAED) patterns (Figure 4 (b), (d), (f) and (h)) of all the  $\text{In}_2\text{O}_3$  samples show spotty ring patterns without any additional diffraction spots and rings of second phases revealing their crystalline cubic structure which is in agreement with the XRD result and standard data (JCPDS:71-2195). Increase in calcination temperature results in stronger spotty patterns and the  $\text{In}_2\text{O}_3$  samples calcined at 550 and 600°C show strong spotty patterns, indicating large particle size of highly crystalline cubic structure. The measured interplanar spacings ( $d_{\text{hkl}}$ ) from selected-area electron diffraction patterns in Figure 4 are in good agreement with the values from the standard data (JCPDS:71-2195) as summarized in Table 2.

The high-resolution TEM (HRTEM) image shown in Figure 5 indicates the presence of a lattice fringe in nanoparticles of  $\text{In}_2\text{O}_3$  sample calcined at 550°C. Lattice distances of 0.5021, 0.3483 and 0.2763 nm indicated in

**Figure 3.** FTIR spectra of the  $\text{In}_2\text{O}_3$  nanoparticles calcined at different temperatures for 2 h.



**Figure 4.** TEM images of  $\text{In}_2\text{O}_3$  nanoparticles calcined at (a) 450°C, (c) 500°C, (e) 550°C, (g) 600°C, and bcc-structured polycrystalline SAED pattern of  $\text{In}_2\text{O}_3$  nanoparticles calcined at (b) 450°C, (d) 500°C, (f) 550°C and (h) 600°C.

**Table 2.** Interplanar spacings ( $d_{hkl}$ ) of  $\text{In}_2\text{O}_3$  samples calculated from TEM selected-area electron diffraction patterns in Figure 4 compared with the reference values in the standard data (JCPDS:71-2195).

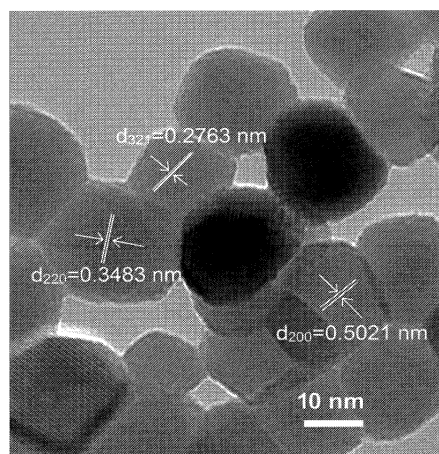
Ring	Calculated interplanar spacing ( $d_{hkl}$ ) $\text{In}_2\text{O}_3$ sample (Å)				Standard data (JCPDS:71-2195)	
	Calcined at 450°C	Calcined at 500°C	Calcined at 550°C	Calcined at 600°C	$d_{hkl}$ (Å)	hkl
1	4.006	4.059	3.944	4.070	4.130	211
2	2.837	2.864	2.863	2.857	2.920	222
3	2.471	2.491	2.483	2.481	2.529	400
4	2.330	2.334	2.326	2.344	2.384	411
5	2.117	2.139	2.107	2.113	2.157	332
6	1.930	1.949	1.939	1.943	1.984	431
7	1.748	1.766	1.759	1.755	1.788	440
8	1.604	1.616	1.610	1.619	1.641	611

these figures correspond to the (200) (220) and (321) crystalline planes of  $\text{In}_2\text{O}_3$ , respectively. These are in agreement with the standard data (JCPDS: 71-2195).

Optical properties of  $\text{In}_2\text{O}_3$  nanoparticles were characterized using a UV–Vis spectrophotometer.  $\text{In}_2\text{O}_3$  (Figure 6) shows a strong absorption below 350 nm (3.54 eV) with a well-defined absorbance peak at around 285 nm (4.35 eV). The band gap can be determined by fitting the absorption data to the direct transition equation by extrapolation of the linear portions of the curves so the absorption equals zero (the inset of Figure 6):

$$\alpha h\nu = A(h\nu - E_g)^{1/2} \quad (2)$$

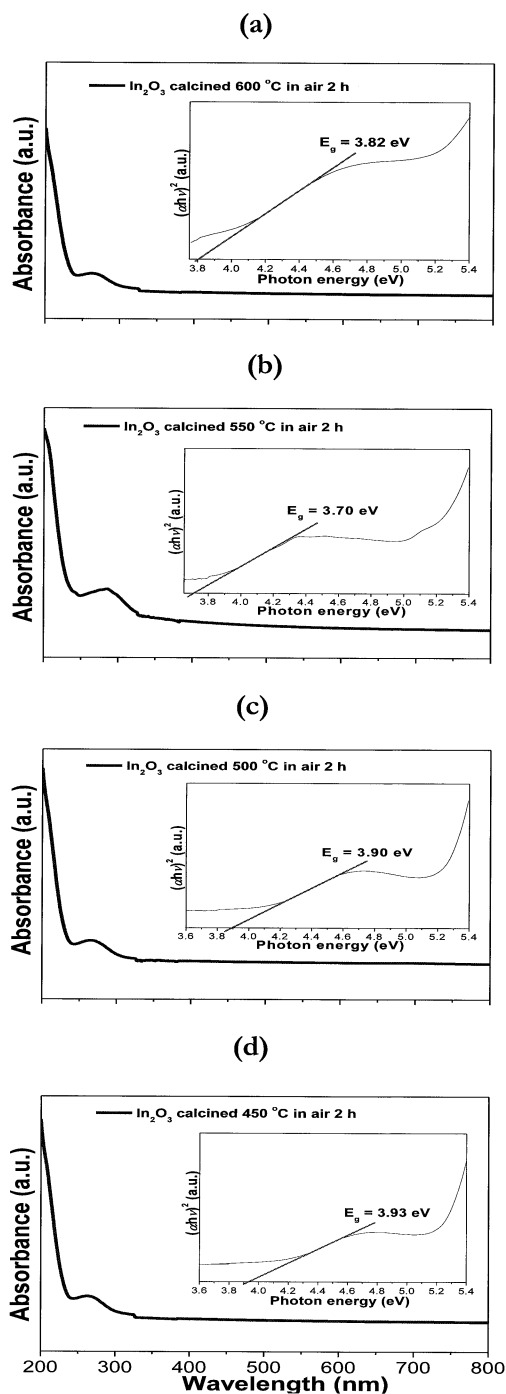
Where  $\alpha$  is the optical absorption coefficient,  $h\nu$  is the photon energy,  $E_g$  is the direct band gap, and  $E_D$  is a constant [26]. An extrapolation of the linear region of a plot of  $(\alpha h\nu)^2$  on the Y-axis versus photon energy ( $h\nu$ ) on the X-axis gives the value of the optical band gap ( $E_g$ ). The plots of  $(\alpha h\nu)^2$  and  $h\nu$  are shown in Figure 6



**Figure 5.** HRTEM image of  $\text{In}_2\text{O}_3$  nanoparticles calcined at 550°C.

for  $\text{In}_2\text{O}_3$  nanoparticles. The calculated values of the optical band gaps for  $\text{In}_2\text{O}_3$  nanoparticles are ~3.93 eV, ~3.90 eV, ~3.70 eV and ~3.82 eV for the  $\text{In}_2\text{O}_3$  samples calcined at 450, 500, 550 and 600°C, respectively. This value is higher than the bulk  $\text{In}_2\text{O}_3$  value of 3.75 eV.

The increase in the optical band gap with increasing temperature may be due to the increase



**Figure 6.** Room temperature optical absorbance spectra of  $\text{In}_2\text{O}_3$  samples calcined in air for 2 h at (a) 450°C, (b) 500°C, (c) 550°C and (d) 600°C. The insets show plots of  $(\alpha h\nu)^2$  as a function of photon energy.

in grain size, the reduction in the disorder and decrease in density of defect states [27-28]. The increase in the optical band gap could also be discussed on the basis of density of state model proposed by Mott and Davis [29]. The quantum size effects have been observed in  $\text{In}_2\text{O}_3$  systems reported in the literature. For example, Maensiri et al. [30] reported band gaps of 30 and 10 nm for  $\text{In}_2\text{O}_3$  nanoparticles prepared using reverse micelles to be 3.29 and 3.31 eV, respectively. Bagheri-Mohagheghi et al. [31] reported band gaps of 10 and 45 nm for  $\text{In}_2\text{O}_3$  nanoparticles synthesized by the sol-gel method to be 4.32 and 4.34 eV, respectively. Since the size of the particles decreases, and the atomic wave functions are overlapping more, we may attribute this to approaching the quantum confinement limit of nano-particles. The quantum confinement effect is expected for semiconducting nanoparticles, and the absorption edge will be shifted to a shorter wavelength (higher energy) when the particle size decreases.

#### 4. CONCLUSION

We have synthesized nanoparticles of  $\text{In}_2\text{O}_3$  by a simple method using indium (III) nitrate and polyvinyl polyhedron (PVP) as the starting chemicals. Structural, morphological and optical properties of the synthesized nanoparticles were characterized. XRD and TEM analysis showed that the  $\text{In}_2\text{O}_3$  samples exhibited a single phase polycrystalline nature with particle sizes of 12 – 25 nm. The morphology and size of  $\text{In}_2\text{O}_3$  materials were affected by the calcination temperature. The optical band gap of synthesized powders exhibited a direct band gap ( $E_g$ ) which was determined to be 3.70-3.93 eV.

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